Magnetization induced local electric dipoles and multiferroic properties of Ba₂CoGe₂O₇

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Abstract

 $Ba_2CoGe_2O_7$, crystallizing in the noncentrosymmetric but nonpolar $P\overline{4}2_1m$ structure, belongs to a special class of multiferroic materials, whose properties are predetermined by the availability of the rotoinversion symmetry. Unlike inversion, the rotoinversion symmetry can be easily destroyed by the magnetization. Moreover, due to specific structural pattern in the xy plane, in which the magnetic Co²⁺ ions are separated by the nonmagnetic GeO₄ tetraherda, the magnetic structure of Ba₂CoGe₂O₇ is relatively soft. Altogether, this leads to the rich variety of multiferroic properties of Ba₂CoGe₂O₇, where the magnetic structure can be easily deformed by the magnetic field, inducing the net electric polarization in the direction depending on the magnetic symmetry of the system, which itself depends on the direction of the magnetic field. In this paper, we show that all these properties can be successfully explained on the basis of realistic low-energy model, derived from the first-principles electronic structure calculations for the magnetically active Co 3d bands, and the Berry-phase theory of electronic polarization. Particularly, we argue that the magnetization induced electric polarization in Ba₂CoGe₂O₇ is essentially local and expressed via the expectation values $\langle \hat{p} \rangle = \text{Tr}[\hat{p}\hat{\mathcal{D}}]$ of some dipole matrices \hat{p} , calculated in the Wannier basis of the model, and the site-diagonal density matrices $\hat{\mathcal{D}}$ of the magnetic Co sites. Thus, the basic aspects of the behavior of Ba₂CoGe₂O₇ can be understood already in the atomic limit, where both magnetic anisotropy and magnetoelectric coupling are specified by $\hat{\mathcal{D}}$. Then, the macroscopic polarization can be found as a superposition of $\langle \hat{p} \rangle$ of the individual Co sites. We discuss the behavior of interatomic magnetic interactions, main contributions to the magnetocrystalline anisotropy and the spin canting in the xy plane, as well as the similarities and differences of the proposed picture from the phenomenological model of spin-dependent p-d hybridization.

I. INTRODUCTION

Magnetically driven ferroelectricity is one of the major topics in the condensed matter physics today. Even though the basic crystallographic symmetry of a magnetic material may not allow for the existence of a spontaneous electric polarization, this symmetry can be further lowered by the magnetic order, which in some cases gives rise to the ferroelectric (FE) activity. After discovery of such effect in TbMnO₃, where the inversion symmetry is broken by some complex noncollinear magnetic order, there is a large number of experimental and theoretical proposals pointing at the existence of a similar effect in other magnetic materials, which are called multiferroics.²

Ba₂CoGe₂O₇ and related compounds take a special place among multiferroic materials. They crystallize in the noncentrosymmetric $P\overline{4}2_1m$ structure,³ which nonetheless does not permit the FE effect because of the fourfold rotoinversion symmetry. Nevertheless, unlike inversion, the rotoinversion symmetry can be relatively easy destroyed by the magnetic order, by canting the magnetic moments out of the rotoinversion axis. Thus, one unique aspect of Ba₂CoGe₂O₇ is that the ferroelectricity in this compound can be induced relatively simple C-type antiferromagnetic (AFM) order. Furthermore, the magnetic structure of Ba₂CoGe₂O₇ is relatively soft: It consists of the CoO₄ tetrahedra, which are interconnected with each other via nonmagnetic GeO₄ tetrahedra (see Fig. 1a). Since two magnetic Co sites in this structure are separated by the long Co-O-Ge-O-Co paths, the exchange interactions between them are expected to be small and the magnetic structure can be easily deformed by the external magnetic field. The possibilities of easy manipulation by the magnetic structure and switching the FE properties have attracted a great deal of attention to Ba₂CoGe₂O₇, and today it was demonstrated in many details how the electric polarization in Ba₂CoGe₂O₇ can be tuned by the magnetic field (Refs. 4 and 5) as well as the uniaxial stress (Ref. 6).

The experimental behavior of the electric polarization in Ba₂CoGe₂O₇ is frequently interpreted basing on the model model of spin-dependent p-d hybridization: if \boldsymbol{e} is the direction of the spin magnetic moment at the Co site, located in the origin, and \boldsymbol{v} is the position of a ligand oxygen atom, relative to this origin, the bond Co-O contributes to the local electric dipole moment as $\boldsymbol{p} \propto (\boldsymbol{e} \cdot \boldsymbol{v})^2 \boldsymbol{v}$. This simple phenomenological expression is able to capture the symmetry properties of the electric polarization and, thus, explain the behavior of this polarization on a phenomenological level. It is therefore not surprising that a

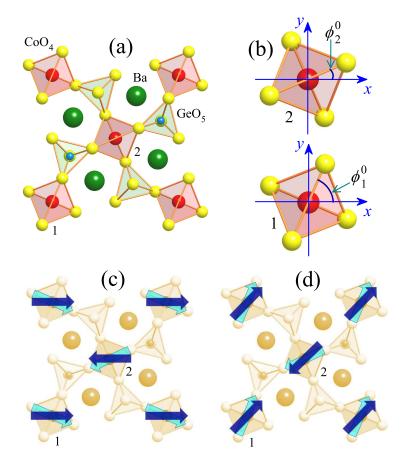


FIG. 1. (Color online) Crystal and spin magnetic structure of Ba₂CoGe₂O₇: (a) General view on the crystal structure in the xy plane. The Co and O atoms are indicated by the medium red and yellow spheres, respectively, the Ba atoms are indicated by the big green spheres, and the Ge atoms are indicated by the small blue spheres. The MnO₄ and GeO₄ tetrahedra are colored red and green, respectively. (b) Details of the rotations of MnO₄ tetrahedra associated with two Mn sites in the xy plane. (c) and (d) Two possible spin magnetic structures, realized without external magnetic field. The light (cyan) arrows indicate the directions of spins favored by the single-ion anisotropy, while the dark (blue) arrows are the joint effect of the single-ion anisotropy and interatomic exchange interactions.

qualitatively similar behavior of the electric polarization was obtained in the first-principles electronic structure calculations,⁸ which generally support the model of spin-dependent p-d hybridization.

On the other hand, the correct quantum-mechanical definition of the electric polarization

is solids should be based on the Berry-phase theory. This theory provides not only an efficient computational framework, which is used today in most of the first-principles electronic structure calculations, but also appears to be a good starting point for the construction of realistic microscopic models, explaining the origin and basic aspects of the behavior of electric polarization in various types of compounds. 10-12 In this work we will pursue this strategy for the analysis of electric polarization in Ba₂CoGe₂O₇. First, we will show that, in the atomic (Wannier) basis, the electric polarization consists of two parts: the local one, which is expressed via the expectation value $\langle \hat{\boldsymbol{p}} \rangle = \text{Tr}[\hat{\boldsymbol{p}}\mathcal{D}]$ of some local dipole matrices $\hat{\boldsymbol{p}}$ and site-diagonal density matrices $\hat{\mathcal{D}}$, and the anomalous one, which is solely related to the phases of the coefficients of the expansion of the wavefunctions in the basis of atomic Wannier orbitals. Similar classification holds for the orbital magnetization.¹³ Then, if the crystal structure possesses the inversion symmetry, the local term becomes inactive and the electric polarization, induced by the magnetic inversion symmetry breaking, is anomalous in origin. Such situation occurs, for instance, in multiferroic manganites with the orthorhombic structure. 11,14 In Ba₂CoGe₂O₇, however, the situation is exactly the opposite: to a good approximation, the electric polarization is expressed as a sum of local electric dipole moments of individual Co sites, while the anomalous contribution is negligibly small. This explains many experimental details of the behavior of electric polarization in Ba₂CoGe₂O₇ as well as basic difference of this compound from other multiferroic materials.

The rest of the paper is organized as follows. The method is described in Sec. II. We use the same strategy as in our previous works, devoted to the analysis of multiferroic properties of various transition-metal oxides on the basis of effective realistic low-energy models, derived from the first-principles calculations. ^{11,14} Therefore, in Sec. II A we will remind the main details of the construction of such effective low-energy model. In Sec. II B, we will analyze the main contributions to the electric polarization in the case of basis, when the Bloch wavefunction is expanded over some atomic Wannier orbitals, centered at the magnetic sites. Since we do not consider the magnetostriction effects, which can low the original $P\overline{4}2_1m$ symmetry of the crystal structure, the considered electric polarization is in fact the electronic one. ⁹ In Sec. III we will present results of our calculations for Ba₂CoGe₂O₇, starting from the atomic limit and, then, consecutively considering the effect of interatomic exchange interactions and the external magnetic field. Finally, in Sec. IV, we will summarize our work. In the Appendix we will estimate the ferromagnetic (FM) contribution to the

interatomic exchange interaction cause by the magnetic polarization of the oxygen 2p band and discuss how such effect can be evaluated starting from the low-energy electron model for the magnetic Co 3d bands.

II. METHOD

A. Effective low-energy model

In this section, we briefly remind the reader the main ideas behind the construction of the effective low-energy electron model. The details can be found in the review article (Ref. 15). The model Hamiltonian,

$$\hat{\mathcal{H}} = \sum_{ij} \sum_{\alpha\beta} t_{ij}^{\alpha\beta} \hat{c}_{i\alpha}^{\dagger} \hat{c}_{j\beta} + \frac{1}{2} \sum_{i} \sum_{\alpha\beta\gamma\delta} U_{\alpha\beta\gamma\delta}^{i} \hat{c}_{i\alpha}^{\dagger} \hat{c}_{i\gamma}^{\dagger} \hat{c}_{i\beta} \hat{c}_{i\delta}, \tag{1}$$

is formulated in the basis of Wannier orbitals $\{\phi_{i\alpha}\}$, which are constructed for the magnetically active Co 3d bands near the Fermi level, starting from the band structure in the local-density approximation (LDA) without spin-orbit (SO) coupling (Fig. 2). Here, each

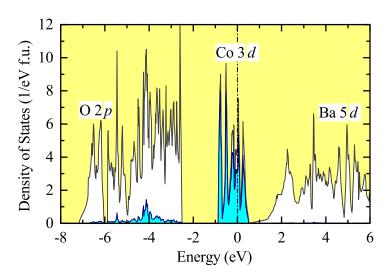


FIG. 2. (Color online) Total and partial densities of states of Ba₂CoGe₂O₇ in the local density approximation. The shaded light (blue) area shows the contribution of the Co 3d states. The positions of the main bands are indicated by symbols. The Fermi level is at zero energy (shown by dot-dashed line).

Greek symbol $(\alpha, \beta, \gamma, \text{ or } \delta)$ stands for the combination of spin $(\sigma_{\alpha}, \sigma_{\beta}, \sigma_{\gamma}, \text{ or } \sigma_{\delta})$ and orbital (a, b, c, or d) indices, for which we adapt the following order: $xy, yz, 3z^2-r^2, zx, x^2-y^2$. Each lattice point i (j) is specified by the position τ (τ') of the atomic site in the primitive cell and the lattice translation \mathbf{R} . Hence, the basis orbitals $\phi_{i\alpha}(\mathbf{r}) \equiv \phi_{\tau\alpha}(\mathbf{r}-\mathbf{R}-\tau)$ are centered in the lattice point $(\mathbf{R}+\tau)$ and labeled by the indices τ and α . Moreover, they satisfies the orthonormality condition:

$$\langle \phi_{\tau'\alpha'}(\mathbf{r} - \mathbf{R}' - \boldsymbol{\tau}') | \phi_{\tau\alpha}(\mathbf{r} - \mathbf{R} - \boldsymbol{\tau}) \rangle = \delta_{\mathbf{R}'\mathbf{R}} \delta_{\tau'\tau} \delta_{\alpha'\alpha}. \tag{2}$$

In our case, the Wannier function were calculated using the projector-operator method (Refs. 15 and 16) and orthonormal linear muffin-tin orbitals (LMTO's, Ref. 17) as the trial wave functions. Typically such procedure allows us to generate well localized Wannier functions, that is guaranteed by the good localization of LMTO's themselves. Then, the one-electron part of the model (1) is identified with the matrix elements of LDA Hamiltonian (\mathcal{H}_{LDA}) in the Wannier basis: $t_{\tau,\tau'+R}^{\alpha\beta} = \langle \phi_{\tau\alpha}(\mathbf{r}-\tau)|\mathcal{H}_{LDA}|\phi_{\tau'\beta}(\mathbf{r}-\mathbf{R}-\tau')\rangle$. Since the Wannier basis is complete in the low-energy part of the spectrum, the construction is exact in the sense that the band structure, obtained from $t_{\tau,\tau'+R}^{\alpha\beta}$, exactly coincides with the one of LDA.

All Wannier basis and one-electron parameters $t_{\tau,\tau'+\mathbf{R}}^{\alpha\beta}$ were first computed without the SO interaction. Then, the SO interaction at each atomic site was included in the "second-variation step", in the basis of the nonrelativistic Wannier functions: $\langle \phi_{i\alpha}(\mathbf{r}) | \Delta \mathcal{H}_{SO} | \phi_{i\beta}(\mathbf{r}) \rangle$, as explained in Ref. 15.

Matrix elements of screened Coulomb interactions can be also computed in the Wannier basis as

$$U_{\alpha\beta\gamma\delta}^{i} = \int d\mathbf{r} \int d\mathbf{r}' \phi_{i\alpha}^{*}(\mathbf{r}) \phi_{i\beta}(\mathbf{r}) v_{\text{scr}}(\mathbf{r}, \mathbf{r}') \phi_{i\gamma}^{*}(\mathbf{r}') \phi_{i\delta}(\mathbf{r}'),$$

where $v_{\rm scr}({\bf r},{\bf r}')$ is obtained using the constrained RPA technique.¹⁸ Then, $v_{\rm scr}({\bf r},{\bf r}')$ does not depend on spin variables and $U^i_{\alpha\beta\gamma\delta} = U^i_{abcd} \, \delta_{\sigma_{\alpha}\sigma_{\beta}} \delta_{\sigma_{\gamma}\sigma_{\delta}}$. Since RPA is very time consuming technique, we employ additional simplifications, which were discussed in Ref. 15. Namely, first we evaluate the screened Coulomb and exchange interactions between atomic Co 3d orbitals, using fast and more suitable for these purposes constrained LDA technique. After that, we consider additional channel of screening caused by the $3d \rightarrow 3d$ transitions in the polarization function of constrained RPA and projecting this function onto atomic 3d

orbitals. The so obtained parameters of screened Coulomb interactions are well consistent with results of full-scale constrained RPA calculations without additional simplifications.

All calculations have been performed for the room temperature $P\overline{4}2_1m$ structure reported in Ref. 3. The corresponding parameters of the low-energy model are summarized in supplemental materials.¹⁹

After the construction, the model is solved in the unrestricted Hartree-Fock (HF) approximation, which is well justified for the considered case where the degeneracy of the ground state is lifted by the crystal distortion.¹⁵

B. Electronic polarization

The electronic polarization can be computed in the reciprocal space, using the formula of King-Smith and Vanderbilt:⁹

$$\mathbf{P} = -\frac{ie}{(2\pi)^3} \sum_{n} \int_{BZ} d\mathbf{k} \langle u_{n\mathbf{k}} | \partial_{\mathbf{k}} u_{n\mathbf{k}} \rangle, \tag{3}$$

where $u_{n\mathbf{k}}(\mathbf{r}) = e^{-i\mathbf{k}\mathbf{r}}\psi_{n\mathbf{k}}(\mathbf{r})$ is the cell-periodic eigenstate of the Hamiltonian $H_{\mathbf{k}} = e^{-i\mathbf{k}\mathbf{r}}He^{i\mathbf{k}\mathbf{r}}$, the summation runs over the occupied bands (n), the **k**-space integration goes over the first Brillouin zone (BZ), and -e (e > 0) is the electron charge. In our case, each $\psi_{n\mathbf{k}}(\mathbf{r})$ is expanded in the basis of Wannier orbitals $\phi_{\tau\alpha}(\mathbf{r} - \mathbf{R} - \boldsymbol{\tau})$, used for the construction of the low-energy model:

$$\psi_{n\mathbf{k}}(\mathbf{r}) = \frac{1}{\sqrt{N}} \sum_{\mathbf{R}\tau\alpha} c_{n\mathbf{k}}^{\tau\alpha} e^{i\mathbf{k}(\mathbf{R}+\boldsymbol{\tau})} \phi_{\tau\alpha}(\mathbf{r} - \mathbf{R} - \boldsymbol{\tau}), \tag{4}$$

where N is the number of primitive cells. Then, the **k**-space gradient of $u_{n\mathbf{k}}$ will have two contributions:

$$\partial_{\mathbf{k}} u_{n\mathbf{k}} = -\frac{i}{\sqrt{N}} \sum_{\mathbf{R}\tau\alpha} (\mathbf{r} - \mathbf{R} - \boldsymbol{\tau}) e^{-i\mathbf{k}(\mathbf{r} - \mathbf{R} - \boldsymbol{\tau})} c_{n\mathbf{k}}^{\tau\alpha} \phi_{\tau\alpha} (\mathbf{r} - \mathbf{R} - \boldsymbol{\tau})$$

$$+ \frac{1}{\sqrt{N}} \sum_{\mathbf{R}\tau\alpha} e^{-i\mathbf{k}(\mathbf{r} - \mathbf{R} - \boldsymbol{\tau})} \partial_{\mathbf{k}} c_{n\mathbf{k}}^{\tau\alpha} \phi_{\tau\alpha} (\mathbf{r} - \mathbf{R} - \boldsymbol{\tau}),$$
(5)

and the electronic polarization **P** will also include two terms:

$$\mathbf{P} = \sum_{n} \int_{BZ} \frac{d\mathbf{k}}{\Omega} \langle c_{n\mathbf{k}} | \hat{\mathbf{p}}_{\mathbf{k}} | c_{n\mathbf{k}} \rangle - \frac{ie}{(2\pi)^3} \sum_{n} \int_{BZ} d\mathbf{k} \langle c_{n\mathbf{k}} | \partial_{\mathbf{k}} c_{n\mathbf{k}} \rangle.$$
 (6)

Here, $|c_{n\mathbf{k}}\rangle$ denotes the column vector $|c_{n\mathbf{k}}\rangle \equiv [c_{n\mathbf{k}}^{\tau\alpha}]$, $\hat{\boldsymbol{p}}_{\mathbf{k}}$ is the matrix $\hat{\boldsymbol{p}}_{\mathbf{k}} \equiv [\boldsymbol{p}_{\mathbf{k}}^{\tau'a',\tau a}\delta_{\sigma_{\alpha'}\sigma_{\alpha}}]$, where

$$\boldsymbol{p}_{\mathbf{k}}^{\tau'a',\tau a} = -\frac{e}{V} \sum_{\mathbf{R}} \langle \phi_{\tau'\alpha'}(\mathbf{r} - \boldsymbol{\tau}') | \mathbf{r} | \phi_{\tau\alpha}(\mathbf{r} - \mathbf{R} - \boldsymbol{\tau}) \rangle e^{i\mathbf{k}(\mathbf{R} + \boldsymbol{\tau} - \boldsymbol{\tau}')}, \tag{7}$$

V is the unit cell volume, and $\Omega = (2\pi)^3/V$ is the first BZ volume. Moreover, since **P** is understood as the *change* of the polarization in the process of adiabatic symmetry lowering (Ref. 9), which in our case is driven by the magnetic degrees of freedom, here and below we drop all nonmagnetic contributions to $\boldsymbol{p}_{\mathbf{k}}^{\tau'a',\tau a}$, which are irrelevant to the magnetic symmetry lowering.

If the transition-metal site is located in the inversion center, the first term of Eq. (6) identically vanishes. Such situation is realized, for instance, in multiferroic manganites, crystallizing in the orthorhombic Pbnm structure, where the FE activity is entirely "anomalous" and associated with the second term of Eq. (6).¹⁴ However, in Ba₂CoGe₂O₇ the situation appears to be exactly the opposite: the first term dominates, while the second contribution is negligibly small (about 0.1 μ C/m² in the magnetic ground state). Furthermore, due to the orthogonality condition (2), the leading contribution to $\mathbf{p}_{\mathbf{k}}^{\tau'a',\tau a}$ comes from the Wannier functions centered at the same atomic site. Then, one can impose in Eq. (7) the additional condition $\mathbf{\tau}' = \mathbf{R} + \mathbf{\tau}$, which yields $\mathbf{p}_{\mathbf{k}}^{\tau'a',\tau a} \approx \mathbf{p}_{\tau}^{a'a}\delta_{\tau'\tau}$, where $\mathbf{p}_{\tau}^{a'a}$ does not depend on \mathbf{k} . Thus, the electronic polarization in Ba₂CoGe₂O₇ will be given by the sum

$$\mathbf{P} \approx \sum_{\tau} \mathbf{P}_{\tau} \tag{8}$$

of the local electric dipoles:

$$\mathbf{P}_{\tau} \equiv \langle \hat{\boldsymbol{p}}_{\tau} \rangle = \text{Tr}_{a} \{ \hat{\mathcal{D}}_{\tau} \hat{\boldsymbol{p}}_{\tau} \}, \tag{9}$$

where Tr_a is the trace over a. Each such dipole is given by the expectation value $\langle \hat{\boldsymbol{p}}_{\tau} \rangle$ of the dipole matrix $\hat{\boldsymbol{p}}_{\tau} = [\boldsymbol{p}_{\tau}^{a'a}]$ and the spin-independent part of the density matrix $\hat{\mathcal{D}}_{\tau} = [\mathcal{D}_{\tau}^{\alpha\alpha'}]$ at the site τ :

$$\mathcal{D}_{\tau}^{aa'} = \sum_{\sigma_{\alpha}} \delta_{\sigma_{\alpha} \sigma_{\alpha'}} \sum_{n} \int_{BZ} \frac{d\mathbf{k}}{\Omega} c_{n\mathbf{k}}^{\tau \alpha} c_{n\mathbf{k}}^{\tau \alpha'*}.$$

In order to evaluate $\hat{\boldsymbol{p}}_{\tau}$ we use the LMTO method and expand each $\phi_{\tau\alpha}(\mathbf{r})$ is the basis $\{\chi_{v\beta}\}$ of linear muffin-tin orbitals, which can be viewed as orthonormalized atomic-like Wannier orbitals, constructed in the whole region of valence states.¹⁷ Moreover, we shift each site τ to the origin, that does not affect the polarization change caused by the magnetic

symmetry lowering. Then, without SO interaction, we have

$$\phi_{\tau a}(\mathbf{r}) = \sum_{vb} q_{\tau a}^{vb} \chi_{vb}(\mathbf{r} - \boldsymbol{v} + \boldsymbol{\tau}). \tag{10}$$

where, $v = \tau$ corresponds to the "head" of the Wannier function, centered at the Co-site τ , while all other contributions describe the "tails" of the Wannier functions, spreading to the neighboring sites v. Then, around each such site we identically present the position operator as $\mathbf{r} = (\boldsymbol{v} - \boldsymbol{\tau}) + (\mathbf{r} - \boldsymbol{v} + \boldsymbol{\tau})$ and assume that the leading contribution to $\hat{\boldsymbol{p}}_{\tau}$ comes from the first term. This is reasonable because, at each site, $(\mathbf{r} - \boldsymbol{v} + \boldsymbol{\tau})$ couples the atomic states with different parity, which are typically well separated in energy. Then, the matrix elements $\boldsymbol{p}_{\tau}^{\alpha\alpha'}$ can be easily evaluated as

$$oldsymbol{p}_{ au}^{a'a} = -rac{e}{V}\sum_{vb}(oldsymbol{v}-oldsymbol{ au})q_{ au a'}^{vb*}q_{ au a}^{vb}.$$

Then, by considering the leading contributions of four oxygen sites surrounding each Co atom, one obtains the following matrices (in μ C/m²):

$$\hat{p}_{1,2x} = \begin{pmatrix} 0 & \mp 3509 & 0 & 6357 & 0 \\ \mp 3509 & 0 & -801 & 0 & -3553 \\ 0 & -801 & 0 & \pm 303 & 0 \\ 6357 & 0 & \pm 303 & 0 & \mp 6142 \\ 0 & -3553 & 0 & \mp 6142 & 0 \end{pmatrix}, \tag{11}$$

$$\hat{p}_{1,2y} = \begin{pmatrix} 0 & 6357 & 0 & \pm 3509 & 0 \\ 6357 & 0 & \mp 303 & 0 & \mp 6142 \\ 0 & \mp 303 & 0 & -801 & 0 \\ \pm 3509 & 0 & -801 & 0 & 3553 \\ 0 & \mp 6142 & 0 & 3553 & 0 \end{pmatrix}, \tag{12}$$

and

$$\hat{p}_{1,2z} = \begin{pmatrix} 0 & 0 & -3682 & 0 & 0 \\ 0 & \pm 3394 & 0 & 3524 & 0 \\ -3682 & 0 & 0 & 0 & \pm 2870 \\ 0 & 3524 & 0 & \mp 3394 & 0 \\ 0 & 0 & \pm 2870 & 0 & 0 \end{pmatrix}, \tag{13}$$

where the upper (lower) signs correspond to the Co-sites 1 (2) in Fig. 1. As expected, the matrices $\mathbf{p}_{1,2} \equiv (\hat{p}_{1,2x}, \hat{p}_{1,2y}, \hat{p}_{1,2z})$ obey the $P\overline{4}2_1m$ symmetry. If the density matrix $\hat{\mathcal{D}}_{1,2}$ obeys the same symmetry, all local electric dipoles will vanish and there will be no net polarization in the ground state. However, if the symmetry of $\mathcal{D}_{1,2}$ is lowered by magnetism, one can expect the appearance of local electric dipoles with some order between sites 1 and 2. If this order is antiferroelectric, there will be no net polarization. Nevertheless, if this order permits a FE component, the system will exhibit a finite net polarization. The details of such symmetry lowering will depend on directions of magnetic moments at the sites 1 and 2. For instance, if the spins are parallel to the z axis, the fourfold rotoinversion around z, \hat{S}_4^z , will remain among the symmetry operations of the magnetic space group, and there will be no local electric dipoles. However, if the moments lie in the xy plane, the rotoinversion \hat{S}_4^z is replaced by the regular twofold rotation, \hat{C}_2^z , which allows for the existence of local electric dipoles parallel to z. For an arbitrary direction of spin, the symmetry will be further lowered, and the local electric dipoles may have all three components. However, the relative directions of dipoles at the sites 1 and 2 will depend on other symmetry operations. Below, we will investigate this magnetic symmetry lowering more in details.

To conclude this section, we would like to emphasize that Eqs. (8) and (9) are the correct quantum-mechanical expressions for the electric polarization in Ba₂CoGe₂O₇, which are based on the very general Berry-phase theory. These expressions allows us clarify the microscopic origin of the polarization, which is frequently ascribed to the spin-dependent p-dhybridization.⁸ The p-d hybridization does play a very important role in this material as it defines the matrix elements of $\hat{\boldsymbol{p}}_{\tau}$ in Eq. (9). Moreover, the contribution of each Co-O bond to the local electric dipole moment is indeed proportional to the vector $(\boldsymbol{v}-\boldsymbol{\tau})$, connecting the Co and O sites. However, the matrix elements $p_{\tau}^{a'a}$ themselves do not depend on the spin state. In this sense, the spin dependence of the hybridization is not the most important aspect. The spin dependence of the electric polarization in our picture comes from the matrix elements of the density matrix, which describes the effect of the SO coupling at the transition-metal sites. The local magnetic moment deforms the electron density around the Co sites. This deformation, which itself depends on the direction of local magnetic moment, spreads to the neighboring oxygen sites via the tails of the Wannier functions and produces finite electric moment. This is the basic microscopic picture underlying the formation of the local electric moments in Ba₂CoGe₂O₇.

III. RESULTS AND DISCUSSIONS

The first important question we need to address is the local properties developed at each of the Co sites. For these purposes we set all transfer integrals in Eq. (1) equal to zero and solve the model in the mean-field approximation separately for each Co site. The SO interaction in these calculations is treated in the frameworks of the self-consistent linear response (SCLR) theory.²⁰ For each direction of spin, it gives us the self-consistent HF potential in the first order of the SO coupling, which can be used again as the input of HF equations in order to obtain at the output the change of the density matrix and the total energy beyond the first order. We would like to emphasize that the considered below effects are beyond the first order of the SO coupling.

The obtained polarization is in the good agreement with the regular self-consistent HF calculations, which can be performed for the high-symmetry points. For the in-plane rotations of the spin magnetization, the results are summarized in Fig. (3). In this case, P_x

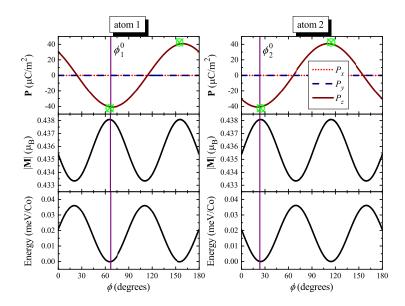


FIG. 3. (Color online) Electronic polarization, absolute value of orbital magnetization, and total energy depending on the direction $e = (\cos \phi, \sin \phi, 0)$ of spin magnetization in the xy plane. The lines show results of self-consistent linear response theory for the spin-orbit coupling.²⁰ The symbols show results of unrestricted Hartree-Fock calculations for the high-symmetry points.

and P_y identically vanish, while $P_z(\phi)$ exhibits the characteristic cosine-like behavior.⁵ As

expected, each $P_z(\phi)$ takes its minimum value at the angle ϕ_{τ}^0 , which specifies the direction of the upper O-O bond of the CoO₄ tetrahedron (see Fig. 1). Note also the minus sign in Eq. (7), which means that the considered polarization is purely electronic. Since two Co sites in Ba₂CoGe₂O₇ are connected by the symmetry operation $\{\hat{C}_2^x|\mathbf{a}_1/2+\mathbf{a}_2/2\}$ (the twofold rotation around x, \hat{C}_2^x , associated with the half of the primitive translations in the xy plane), the angles ϕ_2^0 and ϕ_1^0 satisfy the property: $\phi_2^0 = \pi/2 - \phi_1^0$, where the experimental value of ϕ_1^0 is 67°.³ The maximums of $P_z(\phi)$ are at $\phi = \phi_{\tau}^0 + \pi/2$, which specify the directions of the lower O-O bonds of the CoO₄ tetrahedra. Correspondingly, $P_z(\phi) = 0$ when the spin moment is aligned in between the upper and lower O-O bonds ($\phi = \phi_{\tau}^0 + \pi/4$), and the negative and positive contributions to the polarization cancel each other.

Very generally, the single-ion (SI) anisotropy energy in tetragonal systems has the following form:²¹

$$E_{\rm SI} = K_1 \sin^2 \theta + K_2 \sin^4 \theta + K_2' \sin^4 \theta \cos 4(\phi - \phi_{\tau}^0).$$

By fitting the total energies, obtained in the SCLR calculations, one can find that $K_1 = -2.307 \text{ meV}$, $K_2 = 0.072 \text{ meV}$, and $K_2' = -0.018 \text{ meV}$. The obtained value of K_1 is well consistent with the experimental estimate $K_1 \approx -2.327 \text{ meV}$, reported in Refs. 22 and 23. The parameter K_2' , controlling the in-plane anisotropy, is small but finite. Anyway, this behavior is different from the conventional S = 3/2 spin-only model, where no in-plane anisotropy is expected.²² The difference is caused by an appreciable orbital magnetization ($\sim 0.434 \,\mu_{\rm B}$ per Co site), which is unquenched in the xy plane. Moreover, this orbital magnetization exhibits a sizable anisotropy ($\sim 0.005 \,\mu_{\rm B}$ per Co site – see Fig. 3), depending on the direction in the xy plane. Then, it is reasonable to expect that the magnetocrystalline anisotropy energy should be related to the behavior of the orbital magnetization.²⁴ Indeed, we observe a close correlation between behavior of the SI anisotropy energy and the anisotropy of the orbital magnetization in the xy plane (see Fig. 3). Thus, similar to the polarization, the origin of the in-plane SI anisotropy is related to the orbital degrees of freedom, which become active due to the magnetic symmetry lowering. The SI anisotropy can be further controlled by applying the uniaxial stress.⁶

By summarizing the behavior of the SI anisotropy energy, the spin moments are expected to lie in the xy plane and, at each Co site, be parallel to either upper or lower O-O bond. Therefore, as far as the SI anisotropy energy is concerned, one expects two magnetic configurations, which are depicted in Figs. 1c and 1d. In the first case (Fig. 1c), the spins at the

site 1 and 2 are parallel to, respectively, lower and upper O-O bonds and hence this phase if antiferroelectric. In the second case (Fig. 1d), all spins are parallel to the upper bonds, giving rise to the FE order. In the atomic limit, these two configurations are degenerate.

Another factor, controlling the relative direction of spins, is the interatomic exchange interactions. The isotropic part of these interactions can be computed by considering the infinitesimal rotations of spins, which provides the local mapping of the total energies onto the spin (Heisenberg) model $E_{\rm H} = -\sum_{i>j} J_{ij} \mathbf{e}_i \cdot \mathbf{e}_j$, with \mathbf{e}_i denoting the direction of spin at the site i.²⁵ There are six types of nonvanishing exchange interactions, which are explained in Fig. 4. The values of these interactions, obtained in the C-type AFM state, are

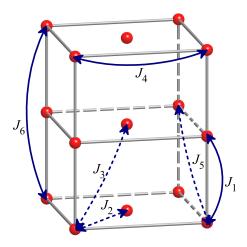


FIG. 4. (Color online) Main parameters of isotropic exchange interactions.

 $J_1 = 0.012$ meV, $J_2 = -2.730$ meV, $J_3 = -0.046$ meV, $J_4 = -0.071$ meV, $J_5 = -0.007$ meV, and $J_6 = -0.012$ meV. Thus, the exchange interaction $J_2 \equiv J$ stabilizes the AFM coupling in the xy plane, while the FM coupling between the planes is stabilized by the combination of J_1 , J_3 , and J_5 . The leading interaction J is relatively weak (compared to other transition-metal oxides), that is directly related with the crystal structure of Ba₂CoGe₂O₇, where neighboring Co sites in the xy plane are separated by relatively long Co-O-Ge-O-Co paths. Nevertheless, the value of J seems to be overestimated in comparison with the experimental data. For example, the Néel temperature $T_N \approx 34$ K, evaluated using Tyablikov's random-phase approximation, 26 is about five time larger than the experimental $T_N = 6.7$ K.³ Similar disagreement is found for the exchange coupling itself: our calculations overestimate the experimental J, reported in Refs. 22, 23, and 27, by the same order of magnitude. This

seems to be a negative aspect of the low-energy model (1), which was constructed only for the Co 3d bands and neglects several important contributions to the magnetic properties of Ba₂CoGe₂O₇, related to the magnetic polarization of the O 2p band. This is also the main reason why we are not able to obtain a good quantitative agreement with the experimental data for the behavior of electric polarization in the magnetic field: although our low-energy model correctly reproduces the main tendencies, the magnetic field should by additionally scaled (by the same factor as the exchange coupling J) to be compared with the experimental data. In the Appendix, we will evaluate the change of the magnetic energy, caused by the polarization of the O 2p band, and show that this mechanism, which favors the FM alignment, can indeed reduce the effective AFM coupling J. Nevertheless, this mechanism overestimates the FM contribution to J and alone does not resolve the quantitative disagreement with the experimental data, which should involve additional factors, such as the exchange striction below T_N .

The Dzyaloshinskii-Moriya (DM) interactions can be calculated by applying SCLR theory for the SO coupling and considering a mixed perturbation, where the SO interaction is combined with rotations of the spin magnetization.²⁰ The DM interactions between nearest neighbors along z are forbidden by the symmetry. Therefore, the strongest interaction is again $\mathbf{d}_2 \equiv (d_x, d_y, d_z)$, which takes place between nearest neighbors in the xy plane. For the bond, connecting the sites 1 and 2 in Fig. 1a, this vector is given by $\mathbf{d}_2 = (-5, 5, -6) \mu \text{eV}$. Similar parameters for other bonds can be obtained by applying the symmetry operations of the space group $P\overline{4}2_1m$, which will change the signs of d_x and d_y . Thus, \mathbf{d}_2 is comparable with K_2 and can also contribute to the canting of spins.

The AFM interaction J enforces the collinear alignment between spins in the xy plane (see Fig. 1), while the SI anisotropy and DM interactions result in a small canting of spins, which is given by:¹⁹

$$\delta\phi \approx \frac{K_2' \cos 4\Phi \sin 4\phi_1^0 + d_z}{2J},\tag{14}$$

where $\Phi = \frac{1}{2}(\phi_1 + \phi_2)$ is the average azimuthal angle, formed by the spins 1 and 2. It is interesting that the magnitude of the canting depends on Φ , which contributes to the SI anisotropy, but not to the DM energy. Indeed, for the antiferromagnetically coupled spins, being parallel to the [100] and [110] axes in the xy plane, $\cos 4\Phi$ is equal to 1 and -1, respectively. Therefore, in the first case (Fig. 1c), the effects of the SI anisotropy and DM interactions will partly compensate each other (note that $\sin 4\phi_1^0 < 0$), while in the second

case (Fig. 1d), these two terms will collaborate, that leads to larger spin canting. This analysis is totally consistent with results of unrestricted HF calculations for the low-energy model (1).

The exchange coupling J, in the combination with the SI anisotropy K'_2 , lifts the degeneracy between states e||[100] and e||[110]. However, the corresponding energy difference (per one formula unit),¹⁹

$$\Delta E = 2K_2' \cos 4\phi_1^0,$$

is very small (about 1 μ eV), mainly because $\cos 4\phi_1^0$ is small. This is the main reason why the direction of spins in the xy plane cannot be not easily determined experimentally.^{22,27}

As was discussed in Sec. II B, the finite value of the polarization is due to the magnetic symmetry lowering, which is reflected in the behavior of the density matrices $\hat{\mathcal{D}}_{1,2}$. The latter can be identically presented as $\hat{\mathcal{D}}_{1,2} = \hat{\mathcal{D}}_{1,2}^0 + \delta \hat{\mathcal{D}}_{1,2}$, where $\hat{\mathcal{D}}_{1,2}^0$ is the average density, obeying the $P\overline{4}2_1m$ symmetry, and $\delta\hat{\mathcal{D}}_{1,2}$ is a perturbation, which depends on the direction of spin e. Straightforward unrestricted HF calculations yield (in 10^{-3})

$$\operatorname{Re}\delta\hat{\mathcal{D}}_{1,2} = \begin{pmatrix} 0 & 0 & \mp 3.92 & 0 & 0\\ 0 & 3.43 & 0 & \mp 1.90 & 0\\ \mp 3.92 & 0 & 0 & 0 & -1.30\\ 0 & \mp 1.90 & 0 & -3.43 & 0\\ 0 & 0 & -1.30 & 0 & 0 \end{pmatrix}$$
(15)

and

$$\operatorname{Re}\delta\hat{\mathcal{D}}_{1,2} = \begin{pmatrix} 0 & 0 & -0.16 & 0 & 0\\ 0 & \pm 1.85 & 0 & -2.88 & 0\\ -0.16 & 0 & 0 & 0 & \mp 3.90\\ 0 & -2.88 & 0 & \mp 1.85 & 0\\ 0 & 0 & \mp 3.90 & 0 & 0 \end{pmatrix}$$
(16)

for the solutions with the spin magnetization being parallel to the axes [100] and [110], respectively (see Figs. 1b and c), where the upper (lower) signs correspond to the Co-sites 1 (2) in Fig. 1. By combining these matrices with $\hat{p}_{1,2z}$, given by Eq. (13), one obtains the following contributions of the sites 1 and 2 to the electric polarization: $P_{1,2z} = \pm 31.2 \,\mu\text{C/m}^2$ and $P_{1,2z} = -28.9 \,\mu\text{C/m}^2$ for e||[100] and e||[110], respectively. Thus, the magnetic structure with e||[100] preserves the symmetry operation $\{\hat{C}_2^x|a_1/2+a_2/2\}$, connecting the sites 1 and

2, which results in the antiferroelectric state. On the other hand, the magnetic structure with e||[110] breaks this symmetry, giving rise to the FE order. The total polarization $P_z = P_{1z} + P_{2z} = 57.8 \ \mu\text{C/m}^2$ is in fair agreement with the experimental data. According to Eqs. (13), (15), and (16), such a behavior is related to the phases of the matrix elements of $\text{Re}\delta\hat{\mathcal{D}}_{1,2}$, which interplay with the phases of $\hat{p}_{1,2z}$: The phases are organized in such a way that for e||[100] and e||[110] their interplay yields $P_{2z} = P_{1z}$ and $P_{2z} = -P_{1z}$, respectively. Note also that the obtained matrices $\text{Re}\delta\hat{\mathcal{D}}_{1,2}$ do not couple with $\hat{p}_{1,2x}$ and $\hat{p}_{1,2y}$, so that the x and y components of the polarization are identically equal to zero.

Next, we discuss how the electric polarization can be controlled by the external magnetic field H. First, we consider the situation, where the AFM spins are parallel to the [110] axis (Fig. 1d) and apply H along the perpendicular direction [110]. The results are summarized in Fig. 5, where for an easier comparison with experimental data we plot $-P_z$.²⁸ In this case,

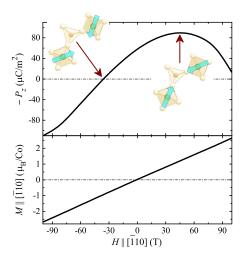


FIG. 5. (Color online) Electric polarization and net spin magnetization in the external magnetic field parallel to the $[\bar{1}10]$ axis, as obtained from the solution of the low-energy electron model in the Hartree-Fock approximation.

H controls the magnitude of the spin canting and the direction of spins relative to the upper (lower) O-O bonds. When both spins are parallel to the upper O-O bonds, $-P_z$ takes the maximal value. The corresponding magnetic field can be easily found from the analysis of the spin Hamiltonian, which yields

$$\mu_{\rm B}H_m = -\frac{8J}{qS}\sin\left(\phi_1^0 - \frac{\pi}{4}\right).$$

Using above values of J and ϕ_1^0 , $g \approx 2$ and S = 3/2, H_m can be evaluated as 47 T, which is in the very good agreement with the results of HF calculations for the low-energy electron model displayed in Fig. 5. Nevertheless, H_m is overestimated by factor five in comparison with the experimental data,^{5,6} following similar overestimation of J, as was explained above.

When the spins are aligned in between the upper and lower O-O bonds, P_z is equal to zero. The maximal value of the polarization at $H = H_m$ is $P_m \approx 90 \ \mu\text{C/m}^2$, which is in fair agreement with the experimental data (about 120 $\mu\text{C/m}^2$, Ref. 5).

Since the in-plane anisotropy is small, the spins in the xy plane can be easily rotated by the magnetic field, which couples to the net magnetization. The results of such calculations are displayed in Fig. 6. In this case, the value of the magnetic field also plays an important

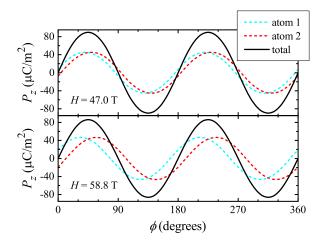


FIG. 6. (Color online) Behavior of electric polarization (total and partial contributions of the Co sites 1 and 2) under the rotation of the external magnetic field in the plane xy, as obtained from the solution of the low-energy electron model in the Hartree-Fock approximation. Here, the angle ϕ specifies the direction of the magnetic field, while the antiferromagnetic component of the magnetization is perpendicular to the filed.

role, as it controls the angle between spins at two Co sublattices: if $H \approx H_m$, the individual contributions P_{1z} and P_{2z} change "in phase", and for $\phi = \pi/4$ (modulo $\pi/2$) $|P_{1z}+P_{2z}|$ reaches its maximal possible value P_m . However, if $H \neq H_m$, there is some "dephasing" and $|P_{1z}+P_{2z}|$ is smaller than P_m .

As was already discussed in Sec. IIB, for an arbitrary direction of spins, the original $P\overline{4}2_1m$ symmetry is completely destroyed and the polarization **P** can also have an arbitrary

direction. Similar to the in-plane rotations (Fig. 3), this behavior can be well understood already in the atomic limit, by considering the change of the density matrix, induced by the SO interaction at a given Co site, which couples with the electric dipole matrix \hat{p}_{τ} . For these purposes we again employ the SCLR method. It is convenient to start with the AFM configuration of spins parallel to the [110] axis and rotate them out of the xy plane. The results of such calculations are summarized in Fig. 7. One can clearly see that, in addition

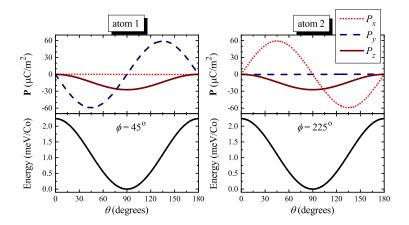


FIG. 7. (Color online) Electronic polarization and total energy depending on the direction $e = (\cos \phi \sin \theta, \sin \phi \sin \theta, \cos \theta)$ of the spin magnetization, rotated out of the xy plane for $\phi = \pi/4$ and $\phi = 5\pi/4$, as obtained in the self-consistent linear response theory for the spin-orbit coupling.²⁰ The azimuthal angles $\phi = \pi/4$ and $\phi = 5\pi/4$ at the sites 1 and 2, respectively, were chosen to simulate the antiferromagnetic spin alignment in the xy plane.

to P_z , there are finite perpendicular components of the polarization, P_x and P_y , which obey certain symmetry rules and replicate the geometry of the rotated CoO₄ tetrahedra. The total energy change in this case is mainly controlled by relatively large anisotropy parameter K_1 .

In practice, such a situation can be realized by applying the magnetic field along the z axis, which cants the spins out of the xy plane (Fig. 8). The angle θ between the spins and the xy plane can be estimated as $\cos \theta = -\mu_{\rm B}gHS/(8J+2K_1)$. There is a clear similarity with the atomic picture, depicted in Fig. 7: the magnetic field slightly decreases $|P_z|$ and induces two perpendicular components of the polarization, which satisfy the condition $P_y = -P_x$. According to the atomic calculations (Fig. 7), the components P_y and P_x are mainly induced at the Co sites 1 and 2, respectively, that is closely related to the geometry of the rotated CoO₄ tetrahedra.

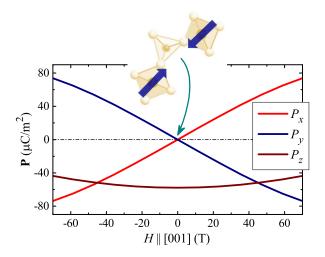


FIG. 8. (Color online) Electric polarization in the external magnetic field parallel to the z axis, as obtained from the solution of the low-energy electron model in the Hartree-Fock approximation. The starting configuration, where antiferromagnetic spins were parallel to the [110] axis, is shown in the inset.

IV. SUMMARY AND CONCLUSIONS

Using effective low-energy model, derived from the first-principles electronic structure calculations, we have investigated the multiferroic properties of Ba₂CoGe₂O₇. There are two important aspects, which make this material especially interesting in the field of multiferroic applications: (i) Ba₂CoGe₂O₇ crystallizes in the noncentrosymmetric but nonpolar structure. The key symmetry operation, which controls the multiferroic properties of this material is the rotoinversion. From the view point of magnetic symmetry breaking, there is a fundamental difference between inversion and rotoinversion. The magnetic pattern, which can break the inversion symmetry, should be rather nontrivial. Typically, such magnetic order arises from the competition of many magnetic interactions and crucially depends on a delicate balance between these interactions. On the contrary, the rotoinversion symmetry can be broken very easily, simply by caning the magnetic moments out of the rotoinversion axis. (ii) The magnetic structure of Ba₂CoGe₂O₇ is very soft and can be easily deformed by the external magnetic field. This property is related to the specific geometry of the Ba₂CoGe₂O₇ lattice, where the CoO₄ tetrahedra are interconnected with each other via the GeO₄ octahedra. Thus, the magnetic Co atoms are separated by the long Co-O-Ge-O-Co paths, resulting

in the relatively weak exchange coupling J. Another important ingredient is the weak inplane anisotropy, which is inherent to magnetic compounds with the spin 3/2.²² We propose that this anisotropy has an intraatomic nature and related to the anisotropy of the orbital magnetization.

Nevertheless, it seems that the magnetic softness of $Ba_2CoGe_2O_7$ has also one negative aspect: the Néel temperature $T_N = 6.7$ K, below which the multiferroic behavior has been observed, is relatively small.³ This imposes a serious constraint on the practical realization of the considered effects: since T_N is controlled by the same exchange coupling J, any attempts to increase T_N will make it more difficult to deform the magnetic structure and manipulate the properties of $Ba_2CoGe_2O_7$ by the magnetic field.

On the theoretical side, we have shown that the electric polarization of Ba₂CoGe₂O₇ can be presented as the sum of electric dipoles, which are induced at each Co site by the local exchange field. Each such dipole is given by the expectation value $\langle \hat{\boldsymbol{p}} \rangle = \text{Tr}[\hat{\boldsymbol{p}}\hat{\mathcal{D}}]$ of the dipole matrix $\hat{\boldsymbol{p}}$ and the site-diagonal density matrix $\hat{\mathcal{D}}$. This is rather general property of Ba₂CoGe₂O₇, which was derived starting from the Berry-phase theory of electric polarization in the Wannier basis. The local character of polarization in the case of Ba₂CoGe₂O₇ is directly related to the rotoinversion symmetry.

The magnetic state dependence of the electric polarization is fully described by the sitediagonal density matrix $\hat{\mathcal{D}}$. Any rotation of the local magnetization from the rotoinversion axis lowers the symmetry of the density matrix $\hat{\mathcal{D}}$ at the Co site and induces the local electric dipole due to the transfer of the weight of the Wannier functions to the neighboring oxygen sites. This transfer is possible due to the p-d hybridization. However, the spin dependence of the hybridization itself does not play an important role. The direction of the electric dipole depends, via $\hat{\mathcal{D}}$, on that of the local magnetization. The total polarization of the crystal is the macroscopic average over the microscopic electric dipoles. This is the basic microscopic picture underlying the multiferroic behavior of Ba₂CoGe₂O₇.

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Appendix: Polarization of oxygen band and interatomic exchange interactions

In this appendix, we evaluate the change of the magnetic energy, caused by the polarization of the O 2p band.

After the solution of the low-energy model, consisting of the Co 3d bands, we expand the basis Wannier functions of the model in the original LMTO basis, Eq. (10), and construct the spin magnetization density $m(\mathbf{r}) = n_{\uparrow}(\mathbf{r}) - n_{\downarrow}(\mathbf{r})$, associated with the Co 3d band. In this appendix, n_{\uparrow} and n_{\downarrow} (v_{\uparrow} and v_{\downarrow}) denote the electron densities (potentials) for the majority (\uparrow) and minority (\downarrow) spin states. $m(\mathbf{r})$ has major contributions at the Co sites as well as some hybridization-induced contribution at the oxygen and other atomic sites. Following the philosophy of the low-energy model, ¹⁵ the interaction of $m(\mathbf{r})$ with the rest of the electronic states can be described in the frameworks of the local-spin-density approximation (LSDA). Therefore, our strategy is to evaluate, within LSDA, the exchange-correlation (xc) field $b(\mathbf{r}) = v_{\downarrow}(\mathbf{r}) - v_{\uparrow}(\mathbf{r})$, which is induced by $m(\mathbf{r})$ and polarizes the O 2p band, and find the self-consistent change of $m(\mathbf{r})$ and $b(\mathbf{r})$, caused by the polarization of the O 2p band. For these purposes, it is convenient to use the SCLR theory.²⁰ For simplicity, let us consider the discrete lattice model and assume that all weights of $m(\mathbf{r})$ are concentrated in the lattice points: $m(\mathbf{r}) = \sum_{v} m_v \delta(\mathbf{r} - \mathbf{v})$, where m_v is the local magnetic moment at the site v. Furthermore, we recall that LSDA is conceptually close to the Stoner model.²⁹ Then, the magnetic part of the xc energy can be approximated as $E_{\rm xc} = -\frac{1}{4} \sum_{v} I_v m_v^2$. In practical calculations, the parameters $\{I_v\}$ can be obtained using the values of intraatomic spin splitting and local magnetic moments. In LMTO, the intraatomic exchange splitting can be conveniently expressed via C-parameters of the centers of gravity of the Co 3d states. 17

Then, by introducing the vector notations $\vec{b} \equiv [b_v]$, and the tensors $\hat{\mathcal{I}} = [I_v \delta_{vv'}]$ and $\hat{\mathcal{R}} = [\mathcal{R}_{vv'}]$, the self-consistent field can be found as

$$\vec{b} = \left[1 + \hat{\mathcal{I}}\hat{\mathcal{R}}\right]^{-1}\vec{b}^0,$$

where $\vec{b}^0 = \hat{\mathcal{I}}\vec{m}$ is the xc field induced by the Co 3d band, and the response tensor is obtained in the first order of the perturbation theory for the wavefunctions, starting from the nonmagnetic band structure in LDA:

$$\mathcal{R}_{vv'} = \sum_{ab} \sum_{n}^{\text{occ}} \sum_{n'}^{\text{unocc}} \sum_{\mathbf{k}}^{\text{BZ}} \left\{ \frac{(C_{n\mathbf{k}}^{va})^* C_{n'\mathbf{k}}^{va} (C_{n'\mathbf{k}}^{v'b})^* C_{n\mathbf{k}}^{v'b}}{\varepsilon_{n\mathbf{k}} - \varepsilon_{n'\mathbf{k}}} + \text{c. c.} \right\}.$$
(A.1)

In these notations, $\{C_{n\mathbf{k}}^{va}\}$ are the coefficients of the expansion of the LDA wave functions over the LMTO basis, $\{\varepsilon_{n\mathbf{k}}\}$ are the LDA eigenvalues, and \mathbf{k} runs over the first Brillouin zone (BZ). Moreover, similar to constrained random-phase approximation for the screened Coulomb interactions (Ref. 18), we have to exclude from the summation in Eq. (A.1) the contributions, where both indexes n and n' belong to the Co 3d band. In the present perturbation theory, such terms describe the change of the magnetization, which is caused by the LSDA potential in the Co 3d band. However, these effects are already taken into account in the low-energy model, where the LSDA part is replaced by a more rigorous unrestricted HF approximation with the screened Coulomb interactions. Therefore, such terms should be excluded at the level of SCLR calculations for the LSDA part. In practical calculations, n runs over the occupied O 2p bands and n' runs over the unoccupied Co 3d bands.

Once \vec{b} is known, the change of \vec{m} , caused by the polarization of the oxygen band, can be found as

$$\delta \vec{m} = -\hat{\mathcal{R}} \vec{b}$$

and corresponding change of the xc-field is $\delta \vec{b} = \hat{\mathcal{I}} \delta \vec{m}$. Since O 2p band is fully occupied, the net change of the magnetic moment is identically equal to zero: $\sum_{v} \delta m_{v} = 0$, irrespectively on the type of the magnetic order. Nevertheless, the individual moments δm_{v} can be finite and contribute to the energy. The corresponding correction to the total energy consists of two parts: $\delta E = \delta E_{\text{Co-O}} + \delta E_{\text{O}}$, where $\delta E_{\text{Co-O}} = -\frac{1}{2}\delta \vec{m}^{T}\hat{\mathcal{I}}\vec{m}$ is the interaction of δm_{v} with the "external" xc field, created by the Co 3d band, and δE_{O} is the energy change caused by $\delta \vec{m}$ in the O 2p band. It also consists of two parts: $\delta E_{\text{O}} = \delta E_{\text{sp}} + \delta E_{\text{dc}}$, where δE_{sp} is the single-particle energy, which can be found in the second order of $\delta \vec{b}$ as $\delta E_{\text{sp}} = \frac{1}{4}\delta \vec{b}^{T}\hat{\mathcal{R}}\vec{b}$, $\delta E_{\text{dc}} = \frac{1}{4}\delta \vec{m}^{T}\hat{\mathcal{I}}\delta \vec{m}$ is the double-counting energy, and $\delta \vec{m}^{T}$ is the row vector, corresponding to the column vector $\delta \vec{m}$. Meanwhile, it is assumed that the magnetic energy of the Co 3d band itself is described by the low-energy model in the HF approximation.

 δE may have different values in the case of the FM and AFM alignment of spins in the xy plane. This difference additionally contribute to interatomic exchange interactions in the plane.

In the $P\overline{4}2_1m$ structure of Ba₂CoGe₂O₇, there are three types of oxygen atoms: O1, O2, and O3, which are located in the Wyckoff positions 2c, 4e, and 8f, respectively.³ The obtained parameters $\{I_v\}$ are 0.98, 1.92, 1.10, and 2.01 eV for Co, O1, O2, and O3, respectively.

The magnetic moments are listed in Table I and the energies are in Table II. As expected,

TABLE I. Local magnetic moments m_v , derived from the low-energy model for the Co 3d band, and the moments δm_v , caused by the polarization of the O 2p band in the ferromagnetic (F) and C-type antiferromagnetic (C) state of Ba₂CoGe₂O₇. All magnetic moments are in μ_B per site and the number of such sites in the unit cell is given in the parentheses.

	F		C	
	m_v	δm_v	m_v	δm_{ψ}
Co (×1)	2.251	0.304	2.245	0.234
O1 (×1)	0.005	-0.004	0	0
O2 (×2)	0.004	-0.004	0	0
O3 (×4)	0.184	-0.073	0.180	-0.057

TABLE II. Magnetic contributions to the energy of interaction between Co 3d and O 2p bands $(\delta E_{\rm Co-O})$, the single particle energy in the O 2p band $(\delta E_{\rm sp})$, the total energy of the 2p band $(\delta E_{\rm O})$, and the total energy $(\delta E = \delta E_{\rm Co-O} + \delta E_{\rm O})$ as obtained for the ferromagnetic (F) and C-type antiferromagnetic (C) states. All energies are in meV per one formula unit.

	F	$^{\mathrm{C}}$
$\delta E_{\mathrm{Co-O}}$	-282.98	-217.67
$\delta E_{ m sp}$	-6.46	-3.24
$\delta E_{ m O}$	27.07	16.91
δE	-255.91	-200.76

the moments m_v are distributed mainly between central Co site and its neighboring sites O3. In the FM state, the total moment is $m_{\rm Co} + m_{\rm O1} + 2m_{\rm O2} + 4m_{\rm O3} = 3~\mu_{\rm B}$, which is totally consistent with the value obtained in the Wannier basis. The moments m_v and δm_v are parallel at the Co sites and antiparallel at the oxygen sites. This tendency is consistent with results of the first-principles calculations and can be explained by the hybridization between Co 3d and O 2p states.³⁰ Therefore, the negative sign of $\delta E_{\rm Co-O}$ is due to the contributions of the Co sites, which are partly compensated by the positive contributions of the oxygen sites. The absolute value of $\delta E_{\rm Co-O}$ is larger for the FM state, mainly because $m_{\rm Co}$ and

 δm_{Co} are larger than those in the C-type AFM state. Thus, the Co-O interaction additionally stabilizes the FM order. Then, since the O 2p band is fully occupied, the contribution δE_{sp} is relatively small. The total contribution of the O 2p band to the magnetic energy is positive. This is because the fully occupied O 2p band itself does not favor the magnetism and any magnetic polarization of this band increases the energy. This also explains why δE_{O} is slightly smaller in the C-type AFM state: the magnetic moments δm_v are smaller and, therefore, the magnetic perturbation of the O 2p band is also smaller. Anyway, this effect is considerably weaker in comparison with the change of $\delta E_{\text{Co-O}}$.

In total, the magnetic polarization of the oxygen band favors the FM alignment of spins in the xy plane. By mapping the energy change δE onto the spin model, which includes only nearest-neighbor interactions in the xy plane, the change of the exchange coupling, caused by the polarization of the oxygen band, can be estimated as $\delta J = \frac{1}{4}(\delta E[C] - \delta E[F]) \approx$ 14 meV. Thus, δJ will indeed compensate the AFM exchange coupling, obtained in the low-energy model for isolated Co 3d bands. However, despite the correct tendency, the obtained change δJ is too large (and would lead to the FM alignment in the xy plane). There may be several reasons for it: (i) The SCLR theory may be to crude for treating the magnetic polarization of the oxygen band (in fact, the perturbation, which is introduced by \vec{b}^0 in the O 2p band is not small); (ii) Some quantitative estimates may change by considering the correct crystal structure below $T_{\rm N}$ (which is not available yet). Particularly, the original $P\overline{4}2_1m$ symmetry can be lowered by the exchange striction effects; (iii) Tp some extent, the correlation interactions in the Co 3d band, beyond the HF approximation, will additionally stabilize the C-type AFM order. The corresponding contribution to the total energy difference between FM and C-type AFM states can be estimated using the secondorder perturbation theory (Ref. 15) as 2 meV per one formula unit. This change alone does not significantly change the FM contribution δJ , caused by the polarization of the O 2p band. However, it can collaborate with other effects, such as the exchange striction.

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